The development of high energy X-ray total scattering method at Beijing Synchrotron Radiation Facility*

Caijuan Shi,^{1,2} Dongliang Yang,^{1,2} Lirong Zheng,¹ Yiming Yang,¹ Fei Zhang,¹ Bingbing Zhang,¹ and Xiaodong Li^{1,†}

¹Beijing Synchrotron Radiation Facility, Institute of High Energy Physics, CAS, Beijing, 100049, P R China

²These authors contributed equally: Caijuan Shi, Dongliang Yang.

As a powerful local structure probe, high energy X-ray total scattering method has been used widely in condensed matter physics, materials science and other fields. With a super-conducting wiggler (SCW) and sagittal focusing monochromator, a high energy X-ray total scattering apparatus has been developed at 3W1 beamline of Beijing Synchrotron Radiation Facility, BSRF, a first-generation synchrotron source. The total scattering apparatus mainly consists of a large two-dimensional flat-panel detector and high-energy X-rays of 50-70 keV, enabling total scattering measurements to be carried out for pair distribution function (PDF) analysis with a Q range of 0.5-25 ${\rm \mathring{A}}^{-1}$. Based on this apparatus, a series of in-situ devices were developed, enabling the acquisition and analysis of data with a decent signal-to-noise ratio under various conditions. Demonstration results, including the observation of perovskite oxides with various A-site doping and bioactive glasses upon annealing, were presented. The practices outlined herein validated that the PDF method is capable of characterizing complex material structure at the atomic level. As synchrotron facilities continue to advance, this work stands as a critical source for fostering collaboration and driving progress in the field of scattering science.

Keywords: High energy X-ray, Total scattering, Beijing Synchrotron Radiation Facility, Pair distribution function

I. INTRODUCTION

Bragg diffraction provides experimental access to the atomic scale, revealing the average long-range order of atomic structures in crystalline materials under the assumption of lattice periodicity. However, relying solely on long-range structural information is insufficient. On the one hand, almost all crystals are imperfect, containing various inevitable defects such as point defects, dislocations, and chemical inhomogeneity [1]. Many properties of crystalline materials are profoundly influenced by these local structural disorders. On the other hand, materials with amorphous characteristics cannot be analyzed using Bragg diffraction, necessitating alternative methods to elucidate their local structural tures [2, 3].

A variety of local structure probes are available, including extended X-ray absorption fine structure (EXAFS) [4] nuclear magnetic resonance (NMR), Raman spectra [5, 6] and others. Among these techniques, the pair distribution function (PDF) derived from the high-energy X-ray total scattering method has garnered significant attention because of its compatibility with in-situ devices and excellent signal-to-noise ratio. The total scattering technique involves measuring the complete diffraction pattern, which encompasses both Bragg and diffuse components. This method collects data across reciprocal space, utilizing high-energy synchrotron X-rays to cover a broad range of momentum transfer (Q). From these total scattering measurements, the weighted probability of finding atoms at specific distances from other atoms, known as the PDF, can be determined.

The resolution of the PDF method is determined by the maximum value Q, $Q_{\rm max}$ ($\Delta r \approx 2\pi/Q_{\rm max}$). $Q_{\rm max}$ is limited by the wavelength of the X-ray, λ , the area of the detector, and the distance from sample to detector, since $Q=4\pi\sin\theta/\lambda$, where 2θ is the scattering angle. It is essential to decrease λ to get larger $Q_{\rm max}$. Expect for the wavelength of incident X-ray, high flux is necessary to detect weak signals at high Q values. This is because the scattering intensity decreases significantly at higher Q ranges, and a high flux ensures sufficient signal-to-noise ratio for accurate data collection and analysis. Meanwhile, a lower background is essential to acquire high-quality data. Therefore, a well-collimated light source and a thin container with minimal signal interference are necessary to achieve optimal experimental conditions.

The advent of synchrotron-based radiation sources has pro-45 vided X-rays with high intensity and short wavelengths, en-46 abling the acquisition of accurate and reliable PDF data. To 47 date, a series of beamlines equipped with total scattering tech-48 niques have been establised, including beamline I15-1 at Di-49 amond Light Source (DLS) [7], beamline Powder Diffrac-50 tion (PD) at the Australian Synchrotron [8], beamline 28-ID-1 at National Synchrotron Light Source II (NSLS-II) [9], 52 beamline BL04B2 at Super Photon ring-8 (SPring-8) [10], ID11 and ID15A of European Synchrotron Radiation Facil-54 ity (ESRF) [11, 12]. Notably, the total scattering method was successfully implemented at Shanghai Synchrotron Radiation Facility (SSRF) 13W beamline [13]. Furthermore, the SSRF phase-II beamline project was fully completed in 2023, with the Ultra-hard X-ray Application Beamline providing total 59 scattering capabilities at energies up to 162 keV [14]. Over 60 the past few decades, the total scattering method has been extensively employed in structural studies of liquids (includ-62 ing melts)[15–18], amorphous materials [15, 19], disordered 63 crystalline materials [20, 21] and so on. It is important to em-64 phasize that the pair distribution function (PDF) has played

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[†] Corresponding author, Xiaodong Li, lixd@ihep.ac.cn

67 systems.

The Beijing Synchrotron Radiation Facility (BSRF) is first-generation synchrotron source, featuring an electron storage ring operating at an energy of 2.5 GeV in dedicated 102 mode. BSRF operates in two modes: dedicated mode and 71 parasitic mode. To optimize the use of the parasitic mode, the original biological macromolecule crystallography station was relocated to the 1W2 straight section, while the 3W1 sta-75 tion was transformed into a high-energy beamline facility. In 76 2019, a superconducting wiggler (SCW) was successfully in-77 stalled in the 3W1 beamline of BSRF, replacing the origi-78 nal permanent magnetic wiggler. This upgrade provides a broader energy range from 40~keV to 80~keV while delivering $_{\text{107}}$ sufficiently high flux. These improvements have motivated us to develop a high-energy X-ray total scattering apparatus at the 3W1 beamline. The aim of this study is to report the opti-83 mization process of this device and to present benchmark total 84 scattering measurements conducted at this beamline, show-85 casing the capabilities it offers to our user community.

II. THEORETICAL BASIS

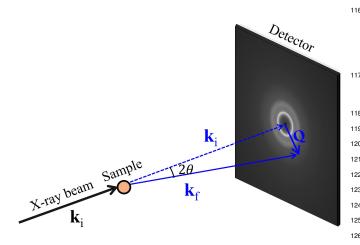


Fig. 1. Scattering triangle of wavevectors of a typical total scattering experiment [22].

88 scattering experiment is shown in fig.1, where ki and kf repre- 132 high Q region can meet the requirement of PDF experiment 89 sent the momentum before scattering and after scattering, re- 133 in many cases. 90 spectively. The scattering intensity includes both elastic scat-91 tering and inelastic scattering, but elastic scattering plays a 92 major role and determines the diffraction pattern. Here we 93 assume that there was no energy exchange between the inci-94 dent quantum and sample, such that $k_i = k_f = 2\pi/\lambda$. Q is 95 the scattering vector, $Q = \mathbf{k}_i - \mathbf{k}_f$, for isotropic samples, |Q| =Q. As such, the scattering vector Q is related to the incident wavelength o and scattering angle 2θ via:

$$|Q| = Q = 2k_i \sin \theta = \frac{4\pi}{\lambda_0} \sin \theta. \tag{1}$$

65 a key role in elucidating local structures, which are funda- 99 In a scattering experiment, the structure factor, S(Q), is re-66 mentally critical for understanding amorphous and disordered 100 lated to the measured differential scattering cross section, 101 I(Q) [23]:

$$S(Q) - 1 = \frac{I(Q) - \left(\sum_{i} c_{i} f_{i}^{2}(Q)\right) - C(Q)}{\left[\sum_{i} c_{i} f_{i}(Q)\right]^{2}}$$
(2)

where c_i is the atomic fraction of element i, $f_i(Q)$ the X-ray atomic form factor, and C(Q) the inelastic (Compton) scat-105 tering contribution. The pair distribution function G(r) could be calculated by Fourier transformation of S(Q) via:

$$G(r) - 1 = \frac{1}{2\pi^2 r \rho} \int_{Q_{\min}}^{Q_{\max}} Q[S(Q) - 1] \sin(Qr) dQ$$
 (3)

where Q_{\min} and Q_{\max} represent the finite range in reciprocal $_{
m 109}$ space used during Fourier Transform, which is limited by Qrange of the instrument, and ρ is the atomic number density in \mathring{A}^3 [24]. Except for G(r), total density function D(r) and total correlation function T(r) is often used in the publica-113 tions [23], $D(r)=4 \pi \rho r[G(r)-1], T(r)=4\pi \rho rG(r)$, peaks in T(r) indicate the existence of atoms with a density exceeding the average number density at a distance r, whereas a 116 valley suggested the absence of atoms.

LIGHT SOURCE PARAMETERS

The total scattering method was designed and developed at 3W1 beamline of BSRF. Currently the 3W1 is the only beamline that can provide high energy (E > 50 keV) X-ray flux in the BSRF with a 2.3 T super-conducting wiggler. A sagittal focusing monochromator with a Si (111) crystal was placed at 32 meters away from the source. With horizontal focusing of the monochromator, the beam size before beam defining slit (slit 1 in fig.3) is 1.7(H) mm \times 2.9(V) mm. Total photon energy is in the range of 50-70 keV and the energy resolution is better than 0.65%. The beam will be optimized by adjusting bending radius of the second crystal before the total scattering experiment. Calculated photon flux and Q space resolution of 130 3W1 beamline are shown in fig.2. From fig.2(b), subtle dif-The scattering triangle of wavevectors in a typical total 131 ferences for $\Delta Q/Q$ at three energy and a value of $\sim 0.3\%$ at

Table 1. Parameters of BSRF 3W1 beamline for total scattering experiment

Animal	Description
Energy (keV)	50-70
Energy resolution	0.65%
Detector	Mercu 1717HS, Mar345
Beam size at sample (mm)	0.8×0.8

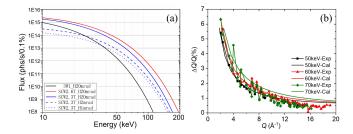


Fig. 2. (a) Calculated photon flux of BSRF 3W1 beamline (blue dashed line); (b) Measured (symbol lines) and calculated (lines) FWHM of selected peaks from the 50 keV, 60 keV and 70 keV total scattering data shown as $\Delta Q/Q$.

APPARATUS OPTIMIZATION PROCESS

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Experimental setup

High energy X-ray total scattering apparatus were designed and conducted in the range 50-70 keV (wavelengths from 0.247-0.178 Å). Obtaining a clean background is quite impor-139 tant before a total scattering experiment, as the background 140 signal need to be subtracted reliably during data processing. Weak background signal is crucial for weak signal samples, e.g., thin films, nano-materials and other disordered systems. Here we utilized several pairs of slits and an evacuated stainless-steel tube with Kapton windows to decrease parasitic scattering and air scattering. Before the experiment, all of the optics were aligned carefully. 146

The first set of slit (labeled A in fig.3) close to the incom-148 ing beam was used to define the beam size and shape. The beam define slit was composed of four tungsten blades with a 149 thickness of 2 mm and the blades position could be adjusted 150 by high precision slide, thus defining the beam size to 0.5-0.8 151 mm. Parasitic scattering produced by the beam define slit was unavoidable, so a clean-up slit (C in fig.3) was designed to clean up slit (F in fig.3) and a big lead shield were placed 191 just in front of the sample to remove the scatter or diffraction 192 160 generated by the devices upstream. 162

and construction, Mar345 detector was utilized and placed 195 target energy; (2) using the standard sample LaB₆ and the imental efficiency and get a larger Q_{max} value, we updated 197 preliminarily; (3) moving the sample station by motor with a the detector to Mercu 1717HS, which is a high speed large 198 known distance; (4) calculating scattering angle 2θ according two-dimensional flat-panel detector and combines a pixelated 199 to the geometrical relation of two position of LaB₆, and the detector with a CsI sensor and features 139 μ m pitch pix- 200 exact energy by using Bragg equation. els, 3072 × 3048-pixel submodule arrangement, generating a 201 172 a sample to detector distance of 180 mm and an energy of 60 203 posure duration of 60 seconds. The X-ray total structure fac- $_{173}$ keV, a Q range of 0.5-25 Å $^{-1}$ could be obtained. It is worth $_{204}$ tor S(Q) and pair distribution functions G(r) are illustrated 174 to mention that benefiting from the high frame rate of Mercu 205 in fig.4a and fig.4b, respectively. The fitting results of CeO₂

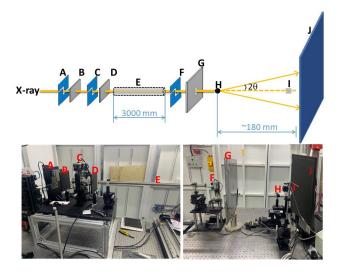


Fig. 3. Schematic layout and Photographs of a typical high-energy scattering set up at 3W1 beamline of BSRF. A: beam define slit (slit 1); B: lead shield 1; C: clean up slit (slit 2); D: lead shield 2; E: evacuated stainless steel tube with mylar windows; F: clean up slit (slit 3); G: lead shield 3; H: sample station; I: beamstop; J: detector

tering and diffraction data up to subseconds, although the data quality might not be good enough due to the low photon flux. 178 XRD pattern of 304 stainless steel during remelting process was collected with a temporal resolution of 0.1 s.

Determination of incident X-ray energy

The most important parameter in PDF experiment, Q_{max} , 182 is limited by the instrument and affected by the energy of the 183 light source. The higher the energy, the larger Q_{max} value that can be achieved, while the scattering factor of X-ray de-185 creases rapidly with the increase of Q value. To obtain large wipe off the parasitic scattering from slit 1. In the meantime, 186 $Q_{
m max}$ and detect the weak signals at high Q region, high entwo lead shields with a thickness of 3 mm were installed just 187 ergy and high flux are usually required. It can be seen that after slit 1 and slit 2 to cut down background radiation. An 188 there is still a high flux at the energy > 50 keV, as shown in evacuated stainless-steel tube with a pressure of 10 Torr was 189 fig.2a and the photon flux decreases with the increase of endesigned to reduce the background from air scatter. Another 190 ergy. Therefore, the data quality at 50 keV was the first to be tested, and then gradually increase the energy to 60 keV and 70 keV.

Calibration of energy was achieved using a 'two distances At the beginning of the total scattering apparatus design 194 method', i.e., (1) adjusting the sagittal monochromator to the 180 mm downstream of the sample. To increase the exper- 196 distance between detector and sample to calibrate the energy

The diffraction patterns of CeO₂ were acquired at three dislarge effective area of 17 × 17 inches. With Mercu 1717HS, 202 tinct energy points, employing a mar 345 detector with an ex-175 1717HS (30 fps), it is possible to attempt to collect total scat- 206 at 60 keV, compared with the standard database reported by

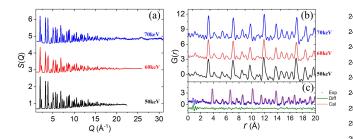


Fig. 4. (a) The X-ray total structure factors S(Q) and (b) pair distribution functions G(r) of CeO_2 at 50 keV, 60 keV and 70 keV with an exposure time of 60 s using a mar345 detector. Note that same result of CeO₂ at 60 keV with comparison to the standard database 257 simulations. reported by Varez et al. [25]

Varez et al [25] were presented in fig.4c. As shown in fig.4c, the peak positions and intensities of G(r) obtained from the BSRF 3W1 beamline aligned well with the standard database, 210 except for the unphysical oscillation in the low r region below 1.5 Å.

The selection of energy points was determined by compar- $_{213}$ ing the comprehensive data quality. The maximum Q value 214 of instrument at 50 keV, 60 keV and 70 keV is 22, 27 and 215 32 Å $^{-1}$, respectively. From the view of $Q_{\rm max}$, 70 keV is a 216 good choice. However, in the high Q range, the oscillation of 70 keV is more severe than 50 keV and 60 keV as the flux $_{\mbox{\scriptsize 218}}$ is lower. From the perspectives of both $Q_{\rm max}$ and the fluc- $^{\mbox{\tiny 258}}$ $_{219}$ tuation of low r region (fig.4b), 60 keV was considered as $_{259}$ results, A-O and Ti-O bond lengths for SrTiO $_3$ -based per-220 the target energy for total scattering experiment. The energy could be adjusted at the range of 50-70 keV according to the 261 requirement of users.

BENCHMARK RESULTS

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A. Ex-situ total scattering experiment of high-entropy perovskite oxides

For ex-situ experiments, samples were collected with Compton tapes and glued to an aluminum alloy frame with an aluminum-tungsten-aluminum 'sandwich' configuration, to suppress stray light. The sample station repeatability is controlled at an accuracy of 0.1 mm by changing samples with a magnetic base. Mercu 1717HS detector was placed 200 mm downstream of the sample. The setup was calibrated using the diffraction pattern of polycrystalline CeO₂ powder. The measurement procedure was controlled by iDetector software and 20 s exposure time was set. Background patterns were 276 collected with the same setup and exposure time. The raw diffraction data were reduced from two-dimensional images 278 structure research. For example, the structural response of and corrected for the effects of polarization and geometry us- 279 glasses to temperature and/or pressure is closely pertinent to ing the program Fit2D [26]. The absorption, geometry, detec- 280 the ionic transport properties, relaxation, and glass-forming 240 tor effect, and the normalization procedure were carried out 281 transition. PDF is one of the most effective means to characusing PDFgetX3 [27]. As a typical perovskite thermoelec- 282 terize the local order and connectivity of glass, so in-situ heat-242 tric oxide, SrTiO₃ displays relatively good electrical proper- 283 ing PDF experiment was used to study the response of differ-243 ties [28]. Total scattering method was utilized to interpret the 284 ent glass structures to temperature. As shown in fig.6a, cou-

244 structure-property relationship of SrTiO₃-based oxides with ²⁴⁵ different A-site doping (La, Ba, Ca, and Pb). The X-ray total structure factors S(Q) and pair distribution functions G(r) of SrTiO₃-based oxides with various A-site doping are shown in fig.5. To explore more detailed structural information, 3D structural models were derived using RMC simulation [29]. The starting configuration consisting of 5000 atoms was created. Several types of constrains were added: the atom-atom approaches, Ti-O connectivity (all titanium atoms were coordinated to certain number oxygen atoms up to 2.5 Å). The choice of these constrains were determined to avoid physically unrealistic structures. Structural information is derived $Q_{\rm max}$ of 22 Å⁻¹ was used during the Fourier Transform. (c) Fitting 256 from counting the atomic configurations generated by RMC

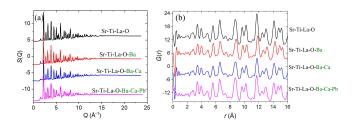


Fig. 5. (a) The X-ray total structure factors S(Q) and (b) pair distribution functions G(r) for SrTiO₃-based perovskite oxides with various A-site doping collected at 3W1 beamline of BSRF.

From the total scattering experiment and RMC simulation ovskite oxides (ABO₃) of different entropies designed at A sites were extracted to calculate the tolerance factor t via equation $t=\frac{\operatorname{length}\left(A-O\right)}{\sqrt{2}\operatorname{length}\left(B-O\right)}$. The tolerance factor is a key parameter in perovskite to reflect the correlation between the relative size relationship of elements and structural symmetry, and the extent of the t value deviating from 1 describes the extent of symmetry breaking [30]. In this work, with the t approaching 1, the mobility recovered, meaning that the tolerance factor helps explain why the A-site entropy engineer-269 ing could enhance the weighted mobility and decouple the 270 carrier-phonon transport by only tuning the average element sizes to tailor the symmetry and Ti displacement, revealing 272 the structural origin of mobility recovery. This work was 273 published in Nature Communications, in cooperation with 274 Qinghua University [31].

B. In-situ heating high energy total scattering experiments

High-temperature experiment is the most common and 277 very important for condensed matter physics and material

286 temperature and high-pressure device was designed with a 329 ity, although the detected solutions are colorless. The scat-₂₈₇ large exit angle $2\theta > 40^{\circ}$, with a temperature up to 873 K and ₃₃₀ tering and Raman spectra data of aqueous M₂SO₄(M= Li, 288 pressure to 40 atm (BEIJING OPERANDO TECHNOLOGY 331 Na, K) solutions were shown in fig.7b and 7c, respectively. 289 CO., LTD).

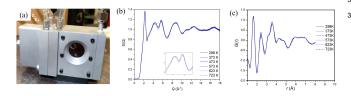


Fig. 6. (a) Photography of the high-temperature and high-pressure device installed at 3W1 beamline, (b) X-ray total structure factors S(Q) and (c) pair distribution functions G(r) of bioactive glasses at different temperatures.

Bioactive glasses are extensively utilized in orthopedics and dentistry. The structural rearrangement upon heating is 292 a key factor in determining its processibility. Here, in cooperation with Shanghai Jiao Tong University, we tracked the structural changes of Hench's composition glass [32] using in-situ high energy X-ray total scattering, and the results are presented in fig.6b and 6c. As shown in fig.6b, the first sharp diffraction peak (FSDP) exhibits minimal variation upon heating, indicating that the medium-range order Si-O-Si rings/chains maintain approximate rigidity and cannot be depolymerized by heating below Tg. This subtle change of FSDP is also reminiscent in real-space structural correlations (as depicted by G(r) plot in the (c) panel). The shoulder peak around 3.2 Å remain almost constant, implying that the Si-Si packing is intact, thereby suggesting that these large Si-O-Si rings or chains are rigid. In contrast to FSDP, the second peak in S(Q) patterns undergoes considerable change with an increase in temperature, indicating the nonnegligible rearrangement of short-range order (SRO) local structures upon 349 heating. The changes of the SRO structures can be intuitively 350 of 2 T (fig.8a) was developed at 3W1 beamline to probe visualized in G(r) patterns, as the intensities of peaks at 2.4 Å and 3.6 Å decreased markedly at high-temperature con- 352 ditions, elucidating that the modifiers-oxygen and modifiers- 353 cations interactions will be much weakened upon heating.

Total scattering coupled with Raman spectra and acoustic 357 314 levitator 315

X-ray total scattering can realize the atomic scale observa-316 317 tion and Raman spectroscopy is sensitive to chemical species 361 G(r) and S(Q) were observed, indicating the weak variation 318 in the solution. To obtain the chemical species informa- 362 of Ga-based alloys microstructure. This apparatus provides a tion and local atomic structural evolution of solutions under 363 powerful method to probe the microstructure of liquid metals low gravity conditions, a multi-mode platform integrating an 364 at various magnetic field intensities. acoustic levitator, portable Raman spectrometer, and high en- 365 the multi-mode platform was shown in fig.7a. In the process 368 800 mm. A Gatan microtension tester equipped with a home-₃₂₆ mm³ droplet, the X-ray beam and Raman spectrometer was ₃₇₀ situ tensile frame, we investigated the anisotropic phase tran-

285 pled with high energy synchrotron radiation facility, a high- 328 droplets were shown in the inset of Fig.7b for visual clar-332 This method is expected to be used in the study of precious samples (1.0 L) and in-situ observation of chemical reactions 334 in solution. It is noteworthy that, combined with laser heat-

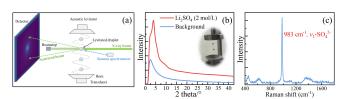


Fig. 7. (a) Schematic of the main components of an acoustic levitator; (b) total scattering patterns with an exposure time of 30 s and (c) Raman spectrum of 2 mol/L Li₂SO₄ aqueous solution droplet, the inset shows several levitated droplets.

337 ing and temperature-controlled system, the levitation device 338 is applicable for detecting the microstructure of high temper-339 ature melts in a container-free environment, eliminating het-340 erogeneous nucleation at the melt-container interface and in-341 creasing the propensity for supercooling. A humility control 342 system is also available, demonstrating the potential accom-343 plishment of in-situ studies on the evaporation and crystalliza-344 tion processes of aqueous droplets. The levitation platform 345 is established in cooperation with Qinghai Institute of Salt 346 Lakes, Chinese Academy of Sciences. More research work is referred to publications by Yongquan Zhou et al [33–35].

Total scattering experiment under in-situ magnetic field

In cooperation with Shanghai University, a magnetic field the microstructure of liquid metals under different magnetic fields and explore the influence mechanism of static magnetic field on Ga-based alloys. During the in-situ magnetic field experiment, samples were injected into capillary tubes and loaded between two magnetic poles, and placed at 200 mm upstream of detector. The structural information was collected using a mar345 detector with an exposure time of 400 s. The X-ray total structure factor S(Q) and pair distribution function G(r) of Ga-based alloys at 0 T and 0.5 T was shown in fig.8b. With the addition of magnetic field, subtle changes of

Apart from total scattering experiment, high energy diffracergy X-ray total scattering was constructed to study the struc- 366 tion experiment coupled with in-situ tensile frame was also ture of levitated aqueous solutions droplets. The schematic of 367 conducted with a longer sample-to-detector distance about of experiment, the solutions were injected into a 0.5 mm³-2 369 built tensile jig was used for tensile testing. Utilizing the in-327 focused on the same droplet simultaneously. Several colored 371 sition, deformation behaviors, orientation evolution of addi-

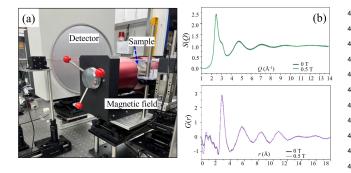


Fig. 8. (a) In-situ magnetic field developed at 3W1 beamline of 416 BSRF; (b) The X-ray total structure factor S(Q) and pair distribution function G(r) of Ga-based alloys at 0 T and 0.5 T.

372 tively manufactured NiTi alloy during loading, in coopera-373 tion with Shanghai University. We refer the reader to relevant publications [36–38], where Pengyue Gao et. al explored 375 the stress-induced phase transformation and lattice correspondence in NiTi shape memory alloy during deformation using in-situ high energy synchrotron X-ray diffraction. In addition, 378 the dislocation density of refractory multi-principal element alloys (MPEAs) under tension was determined according to 380 the Williamson-Hall method, shedding light on the synergis-381 tic combination of ultrahigh strength and large tensile ductil-382 ity in these advanced materials [39–42].

VI. CONCLUSIONS

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386 ing an energy of 60 keV and a large-area detector, a significant amount of high-quality data was obtained and applied in related research. The apparatus is compatible with various experimental environments, including a magnetic field of 2 T, a custom-designed heating furnace with a temperature range of 293–873 K and a pressure of 40 atm, and a tensile frame capable of applying a maximum tension of 2 kN at temperatures 438 ³⁹³ up to 823 K. This work demonstrates that the total scattering apparatus at 3W1 is capable of collecting PDF data with a sat-395 isfactory signal-to-noise ratio for amorphous and disordered 440 399 China.

VII. PERSPECTIVES OF TOTAL SCATTERING TECHNIQUES AT HEPS

402 403 generation synchrotron light sources in Asia, will be accom- 451 energy X-ray experiment.

404 plished by the end of 2025. HEPS will accelerate electrons up 405 to energies of 6 GeV and produce high energy beam that can 406 probe samples at nanometer scales. Its time resolution will be 10,000 times better than that achieved by third-generation synchrotrons [43]. There are a couple of beamlines providing total scattering method in HEPS, e.g., Structural Dynamics, Engineering Materials and High Pressure beamlines. Structural Dynamics Beamline (SDB) of HEPS is dedicated to elucidating the dynamic behavior and structural transformations of materials under varying conditions. Benefiting from the dual advantages of high energy and high flux offered by fourth-generation synchrotron light sources, the SDB beamline provides both conventional PDF and time-resolved PDF 417 methods, utilizing a large-area detector and a direct detector with MHz rates, respectively. Additionally, aerodynamic lev-419 itation techniques will be combined with the PDF method to 420 study the structure of liquids, melts and pharmaceutical drugs under containerless conditions. Engineering Materials Beam-422 line (EMB) provides high energy X-ray beam with an energy 423 of 100 keV, which is proper to conduct typical PDF experi-424 ment with a larger $Q_{
m max}$ and better spatial resolution. High ⁴²⁵ Pressure Beamline (HPB) is featured with high pressure ex-426 periment, where PDF method can be combined with a highpressure technique to study the behavior of systems under ex-428 treme conditions at monochromatic beam between 20 and 75 429 keV with a flux of 10^{12} phs/s. It is worth mentioning that both 430 of Engineering Materials and High Pressure Beamlines pos-431 sess of Eiger2 XE 16M detectors, which will help users col-432 lect diffraction data with better signal-to-noise ratio. At High 433 Pressure and Structural Dynamics Beamline, X-ray beam will 434 be focused to micro scale, holding great promise for detecting 435 solid-liquid interfaces structure during the process of crys-The high energy X-ray total scattering setups developed at 436 tal growth, electro-catalysis and so on. More information 3W1 beamline of BSRF was described in this paper. Utiliz- 437 for beamline information of HEPS can be found online at: 438 https://www.ihep.ac.cn/dkxzz/HEPS/xmgk/HEPSjj/.

VIII. ACKNOWLEDGEMENTS

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- Benmore C J, Gonzalez G B, Alderman O, et al., Hard x-ray 515 [16] methods for studying the structure of amorphous thin films and 516 bulk glassy oxides[J]. J Phys Condens Matter, 2021: 33, 19. 517 doi: 10.1088/1361-648X/abe352
- Gao C, Jiang Z, Wang P, et al. Metal-Organic Framework 519 459 Glass Anode with an Exceptional Cycling-Induced Capac-460 10.26434/CHEMRXIV.14745477.V1 462
- [4] Alderman O, Wilding M C, Tamalonis A, et al. Iron K-edge X- 523 463 ray absorption near-edge structure spectroscopy of aerodynam- 524 [19] 464 ically levitated silicate melts and glasses[J]. Chemical Geology. 525 465 2017, 453: 169-185. doi:10.1016/j.chemgeo.2017.01.020 466
 - [5] Jian, Wang, Jinglin, et al. In-situ studies on the micro-structure 527 [20] evolution of $A_2W_2O_7$ (A = Li, Na, K) during melting by high temperature Raman spectroscopy and density functional the- 529 ory[J]. Spectrochimica Acta Part A: Molecular and Biomolec- 530 ular Spectroscopy. 2017. doi:10.1016/j.saa.2017.05.046
 - Wang, Min, Simon, et al. Quantitative Studies on the 532 [21] Bridges F, Keiber T, Juhas P, et al. Local vibrations and neg-Structure of Molten Binary Potassium Molybdates by in Situ Raman Spectroscopy and Quantum Chemistry 2018. Initio Calculations[J]. Analytical chemistry. doi:10.1021/acs.analchem.8b01470
- Sutter J P, Chater P A, Hillman M R, et al. Three-energy focus- 537 477 ing Laue monochromator for the diamond light source x-ray 538 478 pair distribution function beamline I15-1[J]. AIP Conference 539 479 Proceedings. 2016, 1741(1). doi:10.1063/1.4952877 480
 - D'Angelo A M, Brand H E, Mitchell V D, et al. Total scattering measurements at the Australian Synchrotron 542 Powder Diffraction beamline: capabilities and limitations[J]. 543 [24] Journal of Synchrotron Radiation. 2023, 30(2): 327-339. 544 doi:10.1107/S1600577522011614
- 486 487 X-ray pair distribution function (PDF) beamline and END- 547 station control system[Z]. Brookhaven National Lab.(BNL), 548 488 Upton, NY (United States), 2019. doi:10.18429/JACoW-489 ICALEPCS2019-THBPP04
- 491 [10] Yamada H, Nakada K, Takemoto M, et al. Fully auto- 551 [26] mated measurement system for temperature-dependent X- 552 492 ray total scattering at beamline BL04B2 at SPring-8[J]. 553 493 Journal of Synchrotron Radiation. 2022, 29(2): 549-554. 554 494 doi:10.1107/S1600577521013527 495
- Bernasconi A, Wright J, Harker N. Total scattering experi-496 [11] ments on glass and crystalline materials at the ESRF on the 497 ID11 Beamline[J]. Powder Diffraction. 2015, 30(S1): S2-S8. 498 doi:10.1017/s0885715614001304 499
- [12] Vaughan G B M, Baker R, Barret R, et al. ID15A 560 [28] 500 at the ESRF-a beamline for high speed operando X-ray 501 diffraction, diffraction tomography and total scattering[J]. 502 Journal of Synchrotron Radiation. 2020, 27(2): 515-528. 503 doi:10.1107/S1600577519016813 504
- [13] Xiao-Juan Zhou, Ju-Zhou Tao, Han Guo and He Lin. Atomic 565 pair distribution function method development at the Shanghai 566 506 507 076101. doi:10.1088/1674-1056/26/7/076101 508
- 509 [14] Ren-Zhong Tai1, Zhen-Tang Zhao. Overview of SSRF phase-II 569 beamlines[J]. Nuclear Science and Techniques. 2024, 35: 137. 570 510 511 doi:10.1007/s41365-024-01487-1

- [1] Takeshi Egami, Simon J L Billinge. Underneath the Bragg 512 [15] Shi C, Alderman O L, Berman D, et al. The structure of amorphous and deeply supercooled liquid alumina[J]. Frontiers in Materials. 2019, 6: 38. doi:10.3389/fmats.2019.00038
 - Shi C, Oliver L G A, Tamalonis A, et al. Redox-structure dependence of molten iron oxides[J]. Communications Materials. 2020, 80(1): 1-7. doi:10.1038/s43246-020-00080-4
 - 518 [17] Shi C, Oliver L G A, Tamalonis A, et al. The structure of molten calcium ferrite under various redox conditions[J]. Phil. Trans. R. Soc. A. 2023. doi:10.1098/rsta.2022.0352
 - ity Enhancement for Lithium Ion Batteries[J]. 2021. doi: 521 [18] Wilke S K, Benmore C J, Alderman O L G, et al. Plutonium oxide melt structure and covalency[J]. Nature Materials. 2024. doi:10.1038/s41563-024-01883-3
 - Lan S, Zhu L, Wu Z, et al. A medium-range structure motif linking amorphous and crystalline states[J]. Nature Materials. 2021, 20(10): 1347-1352. doi:10.1038/s41563-021-01011-5
 - Billinge S J L. The rise of the X-ray atomic pair distribution function method: a series of fortunate events[J]. Philosophical transactions of the Royal Society of London. Series A: Mathematical, physical, and engineering sciences. 2019, 377(2147): 20180413. doi:10.1098/rsta.2018.0413
 - ative thermal expansion in ZrW2O8[J]. Phys Rev Lett. 2014, 112(4): 45505. doi:10.1103/PhysRevLett.112.045505
 - 535 Fischer H E, Barnes A C, Salmon P S. Neutron and xray diffraction studies of liquids and glasses[J]. Reports on Progress in Physics. 2006, 69(1): 233. doi:10.1088/0034-4885/69/1/R05
 - Keen D A. A comparison of various commonly used correlation functions for describing total scattering[J]. Journal of Applied Crystallography. 2001, 34(2): doi:10.1107/S0021889800019993
 - Benmore C J. A review of high-energy X-ray diffraction from glasses and liquids[J]. ISRN Materials Science. 2012, 2012. doi:10.5402/2012/852905
- [9] Ivashkevych O, Abeykoon M, J. Adams, G. Bischof Hard 546 [25] Varez, A.; Garcia-Gonzalez, E.; Jolly, J.; Sanz, J. Structural characterization of $Ce_{1-x}Zr_xO_2$ (0 <= x <= 1) samples prepared at 1650 C by solid state reaction. A combined TEM and XRD study[J]. Journal of the European Ceramic Society. 2007, 27: 3677-3682. doi:10.1016/j.jeurceramsoc.2007.02.014
 - Hammersley A P, Svensson S O, Hanfland M, et al. Two-dimensional detector software: from real detector to idealised image or two-theta scan[J]. International Journal of High Pressure Research. 1996, 14(4-6): 235-248. doi:10.1080/08957959608201408
 - Juhas P, Davis T, Farrow C L, et al. PDFgetX3: A rapid and 556 highly automatable program for processing powder diffraction data into total scattering pair distribution functions[J]. J. Appl. Cryst. 2013(46): 560-566. doi:10.1107/S0021889813005190
 - Wang J, Zhang B, Kang H, et al. Record high thermoelectric performance in bulk-SrTiO3 via nano-scale modulation doping[J]. Nano Energy. 2017(35): 395.doi:10.1016/j.nanoen.2017.04.003
 - Mcgreevy R L. Reverse monte carlo modelling[J]. Journal of Physics: Condensed Matter. 2001, 13(46): R877. doi:10.1088/0953-8984/13/46/201
 - Synchrotron Radiation Facility[J]. Chin. Phys. B. 2017, 26(7): 567 [30] Tilley R J D. Perovskites: Structure-Property Relationships[J]. John Wiley Sons. 2016. doi:10.1557/mrs.2017.81
 - Zheng Y, Zhang Q, Shi C, et al. Carrier-phonon decoupling in perovskite thermoelectrics via entropy engineering[J]. Nature Communications. 2024, 7650(15): 1-12. doi:10.1038/s41467-024-52063-5

571

- 573 [32] Cao W, Hench L L. Bioactive Materials[J]. Ceramics Interna-598 [38] Gao P, Zhang Z, Huang J, et al. In-situ synchrotron diffraction tional. 1995, 22: 493-507. doi:0272-8842(95)00126-3 574
- Wang G, Ohara M B K, Ohara K, et al. Hydration of 600 575 [33] Alkali Metal and Halide Ions from Static and Dynamic 601 576 Viewpoints[J]. Journal of physical chemistry letters. 2023. 602 577 doi:10.1021/acs.jpclett.3c01302 578
- [34] Zhou Y, Yamaguchi T, Ikeda K, et al. Dihydrogen Bonds 579 in Aqueous NaBD4 Solution by Neutron and X-ray Diffrac- 605 tion[J]. Journal of Physical Chemistry Letters. 2020, 11(5): 606 [40] 581 1622-1628. doi:10.1021/acs.jpclett.9b03183 582
- [35] Jing Z, Yamaguchi T, Ohara K. Alkali Metal Ion Recogni- 608 583 tion by 18-Crown-6 in Aqueous Solutions: Evidence from Lo- 609 584 cal Structures[J]. The journal of physical chemistry, B. Con- 610 585 densed matter, materials, surfaces, interfaces biophysical. 611 [41] An Z, Yang T, Shi C, et al. Negative enthalpy alloys and local 586 2023, 127(21): 4858-4869. doi:10.1021/acs.jpcb.3c01875
- 588 [36] Gao P, Li R, Shi C, et al. Reveling the orientation prefer- 613 ence along with localized Lüders-type deformation in poly- 614 589 crystalline NiTi SMA by in-situ synchrotron-based high en- 615 [42] An Z, Mao S, Vayyala A, et al. Multiscale hierarchi-590 ergy X-ray diffraction[J]. Vacuum. 2024, 221: 112921. 616 591 doi:10.1016/j.vacuum.2023.112921 592
- 593 [37] Gao P, Li R, He D, et al. Understanding the asymmetric ori- 618 entations and stress states in polycrystalline NiTi SMA by in- 619 situ synchrotron-based high-energy X-ray diffraction[J]. Ma- 620 [43] 595 terials Science and Engineering: A. 2024, 896: 146301. 621 596 doi:10.1016/j.msea.2024.146301 597

- study on the anisotropic deformation and phase transformation behaviors in NiTi shape memory alloy fabricated by laser powder bed fusion[J]. Additive Manufacturing. 2024, 96: 104566. doi:10.1016/j.addma.2024.104566
- [39] An Z, Li A, Mao S, et al. Negative mixing enthalpy solid 603 solutions deliver high strength and ductility[J]. Nature. 2024, 625(7996): 697-702. doi:10.1038/s41586-023-06894-9
 - An Z, Mao S, Liu Y, et al. Inherent and multiple strain hardening imparting synergistic ultrahigh strength and ductility in a low stacking faulted heterogeneous highentropy alloy[J]. Acta Materialia. 2023, 243: doi:10.1016/j.actamat.2022.118516
 - chemical ordering: a concept and route leading to synergy of strength and ductility[J]. National Science Review. 2024, 11: e26. doi:10.1093/nsr/nwae026
 - cal heterostructure yields combined high strength and excellent ductility in a Co-Cr-Fe-Ni-Al negative enthalpy alloy[J]. Acta Materialia. 2024, 281: doi:10.1016/j.actamat.2024.120366
 - Conroy G. World's brightest X-rays: CHINA IS FIRST IN ASIA TO BUILD NEXT-GENERATION SYN-CHROTRON[J]. Nature. 2024, 629: 740. doi:10.1038/d41586-024-01346-4